

3D Cobalt-MOF/CNF Assemblies as Bifunctional Electrocatalysts for Efficient Water Splitting in Alkaline Media

Abstract

The ultrasonication-assisted synthesis of a three-dimensional Cobalt-MOF/Carbon Nanofiber (Co-MOF/CNF) composite has resulted in an electrocatalyst with exceptional structural stability and efficient charge transfer characteristics. The composite exhibits a remarkably low charge transfer resistance (R_{ct}) of 2Ω , indicating superior electron conductivity, while a high double-layer capacitance (C_{dl}) of 12.3 mF/dec enhances its catalytic activity. These properties contribute to its outstanding bifunctional performance in both the hydrogen evolution reaction (HER) and oxygen evolution reaction (OER), achieving low overpotentials of 193 mV and 250 mV , respectively, at a current density of 10 mA/cm^2 . Furthermore, a water-splitting device configured with Co-MOF/CNF electrodes demonstrates efficient overall water electrolysis, requiring only 1.49 V to achieve 10 mA/cm^2 . The system exhibits excellent long-term operational stability, maintaining consistent performance for over 24 hours. These findings highlight the potential of Co-MOF/CNF composites as efficient and durable electrocatalysts for sustainable energy applications.

Introduction:

The recent surge in the consumption and cost of fossil fuels, coupled with their detrimental environmental effects, has underscored the urgent need to explore renewable and sustainable energy sources [1]. Among these, hydrogen has emerged as a promising alternative to conventional fuels due to its eco-friendly nature, high energy density, and ease of storage [2]. Electrochemical water splitting (EWS), encompassing both the oxygen evolution reaction (OER) and hydrogen evolution reaction (HER), represents a viable method for hydrogen production. Nevertheless, this process is hindered by thermodynamic limitations, necessitating the application of additional voltage to overcome these barriers [3].

The mechanisms underlying HER and OER are profoundly influenced by the intrinsic electrochemical and electronic properties of the electrode surface. Enhancing catalytic efficiency requires the optimization of these properties to accelerate sluggish reaction kinetics, which are governed by the catalyst's interaction with water, intermediate species, hydrogen, and oxygen. To facilitate faster reaction kinetics, the development of high-performance

electrocatalytic materials is imperative. Although noble metals and their oxides, such as Pt, RuO₂, and IrO₂, have demonstrated excellent catalytic performance, their limited availability and high cost restrict widespread application [4]. Consequently, research has shifted towards utilizing abundant transition metals as cost-effective alternatives.

Transition metal-based electrocatalysts, including oxides, carbides, nitrides, chalcogenides, and phosphates, have gained considerable attention for their potential in clean hydrogen production through water electrolysis [5]. These materials have shown promise in electrochemical energy conversion and storage devices, such as fuel cells and water electrolyzers, serving as viable substitutes for noble metal-based catalysts [6,7,8].

Metal-organic frameworks (MOFs) have recently attracted significant interest due to their high porosity, tunable pore structures, large surface areas, and favorable electrochemical properties [9]. MOFs have demonstrated potential across diverse applications, including gas capture, storage, catalysis, sensing, and drug delivery [10]. A notable subclass of MOFs is the zeolitic imidazolate frameworks (ZIFs), composed of metal-imidazole-metal linkages with transition metals like nickel (Ni), cobalt (Co), and zinc (Zn). These materials exhibit complex three-dimensional topologies, resembling the structures of aluminosilicate zeolites [11,12,13].

ZIFs combine the advantageous properties of MOFs and zeolites, such as unimodal microporosity, high crystallinity, ultrahigh surface area, exceptional mechanical stability, and customizable functionalities. Despite these benefits, their direct application in electrochemical processes is limited by poor chemical stability and low electronic conductivity. However, ZIFs are promising precursors for developing metal-carbon composite materials suitable for water splitting due to their high carbon and nitrogen content and rich metal ion composition. These features enhance their reactivity, electron transport capabilities, and stability, making them ideal for electrochemical applications [14,15].

Among ZIFs, ZIF-67, composed of Co²⁺ ions and 2-methylimidazole, exhibits cubic crystal symmetry and has garnered significant attention for water splitting applications due to its structural flexibility and the inherent catalytic activity of cobalt [16]. Co-based catalysts are particularly attractive for OER in alkaline media, owing to their cost-effectiveness, environmental compatibility, and abundant availability. As a result, extensive efforts have been directed towards the synthesis, structural optimization, and in-depth understanding of the structure-property relationships of cobalt-based OER catalysts [17].

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However, the poor electronic conductivity of MOFs poses a significant challenge to their overall electrochemical performance. Incorporating carbonaceous materials into MOF structures has proven effective in enhancing their electrochemical efficiency [18]. Various carbon-based materials have been extensively studied as support matrices for active catalytic materials [19,20]. Among these, interwoven hollow one-dimensional (1D) carbon nanofibers (CNFs) offer distinct advantages. Their multichannel hollow architecture facilitates efficient mass and charge transport, thereby accelerating electron transfer processes [21,22]. Additionally, the nanoscale diameter and high aspect ratio of CNFs enable the formation of interconnected three-dimensional networks, promoting mass diffusion and enhancing electrocatalytic performance through improved reaction kinetics [23,24].

In this study, we report the synthesis of a Co-MOF/CNF heterostructure. The Co-MOF was synthesized via a simple solvothermal method using 2-methylimidazole as the organic linker. Subsequently, the Co-MOF was integrated with CNFs through a cost-effective ultrasonication process. The CNFs form a uniform network over the Co-MOF surface, enhancing electrical conductivity and providing additional active sites for ion diffusion, thereby improving electrocatalytic activity. Electrochemical characterization revealed that the Co-MOF/CNF heterostructure functions effectively as a bifunctional catalyst for overall water splitting, exhibiting low overpotentials of 250 mV for OER and 193 mV for HER at a current density of 10 mA/cm².

Experimental Section

Synthesis of Co-MOF:

Co-MOF was synthesized via a solvothermal method by dissolving 0.936 g of Co(NO₃)₂·6H₂O and 2.5 g of 2-methylimidazole in a solvent mixture of ethanol (30 mL), DMF (20 mL), and water (10 mL). The solution was stirred for 30 minutes, transferred to a 100 mL polymer-coated Teflon-lined autoclave, and heated in a hot air oven for 12 hours. After resting for 24 hours, the product was centrifuged with ethanol, and the resulting powder was dried at 60°C overnight.

Synthesis of Co-MOF/CNF Composite:

The Co-MOF/CNF composite was prepared via ultrasonication. CNFs (0.5 mg/mL) were dispersed in ethanol and sonicated for 1 hour, followed by the addition of 1 mg of Co-MOF. The mixture underwent further ultrasonication for 90 minutes, then was washed with water and ethanol, and dried for future use.

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Discussion of results:

The X-ray diffraction (XRD) patterns of Co-MOF and the Co-MOF/CNF composite are presented in Figure 1a. The prominent diffraction peaks for the Co-MOF/CNF composite are observed at 2θ values of 7.1° , 10.2° , 12.5° , 14.5° , 16.4° , 17.9° , 22.0° , 23.3° , 24.5° , 25.5° , and 26.6° , corresponding to the (011), (002), (112), (002), (013), (222), (114), (233), (002), and (134) crystallographic planes, respectively [25]. These diffraction peaks confirm the successful incorporation of Co-MOF within the synthesized composite. Additionally, a low-intensity peak at $2\theta = 25.6^\circ$, indexed to the (002) plane, signifies carbon nanofibers in the composite [26]. The XRD patterns align with previously reported diffraction data, with minor shifts in positions, suggesting the formation of crystalline MOFs on the surface of CNF.

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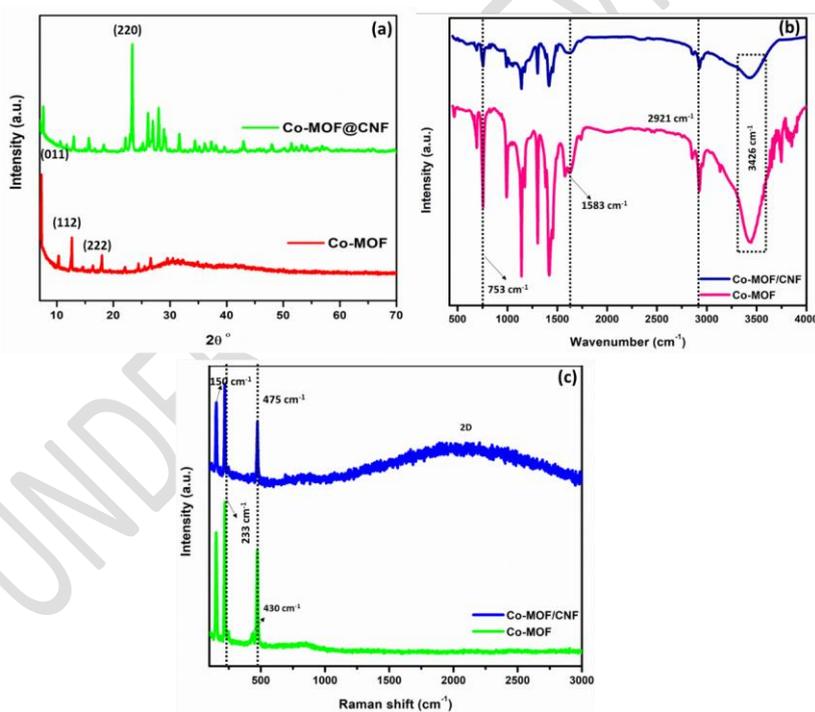


Figure 1: (a) X-ray diffraction patterns, (b) Fourier-transform infrared spectra, and (c) Raman spectra of Co-MOF and Co-MOF/CNF composites

FT-IR spectroscopy was employed to investigate the functional groups present on the surface and to analyze the chemical structural variations among the different catalysts, as illustrated in Figure 1b. The characteristic peak observed in the range of 600–1500 cm^{-1} is attributed to the stretching vibrations of the imidazole ring in Co-MOF. Additionally, the broad absorption band at 3432 cm^{-1} corresponds to the stretching vibrations of N–H and –OH groups [27]. The bending vibration of C=N is responsible for the peak at 753 cm^{-1} , while the absorption bands at 1583 cm^{-1} and 1644 cm^{-1} are assigned to the stretching vibrations of C=N and C=O, respectively [28]. Furthermore, the band at 2921 cm^{-1} indicates C–H stretching vibrations [29]. Variations in the intensity of these peaks suggest effective bonding between CNF and the Co-MOF framework.

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Raman spectroscopy was conducted to further substantiate the successful integration of CNF into the Co-MOF structure at room temperature, as depicted in Figure 1c. The spectra, recorded in the range of 100–3000 cm^{-1} for both Co-MOF and Co-MOF/CNF, display several low-intensity peaks in the lower Raman shift region at 150, 223, 430, and 475 cm^{-1} , which are associated with the presence of Co ions within the Co-MOF structure [30]. The peaks at 430 and 475 cm^{-1} are attributed to Co–O and Co–N (2-methyl imidazole) bonds within the MOF architecture [31]. Additionally, a prominent peak around 2400 cm^{-1} corresponds to second-order defects (D+G bands), indicating the graphitic characteristics of the Co-MOF/CNF composite [32].

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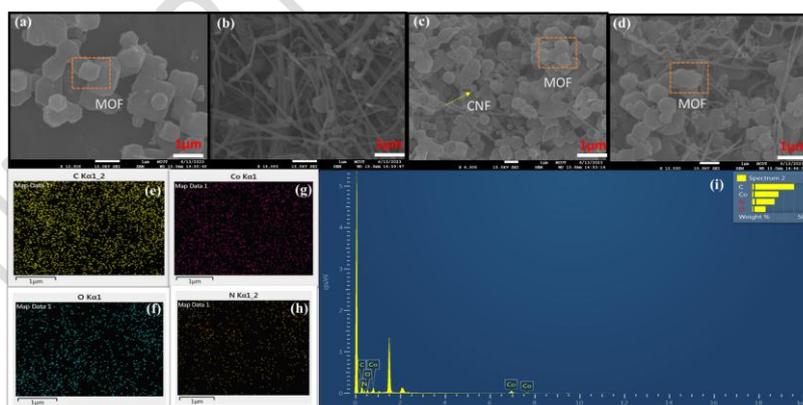
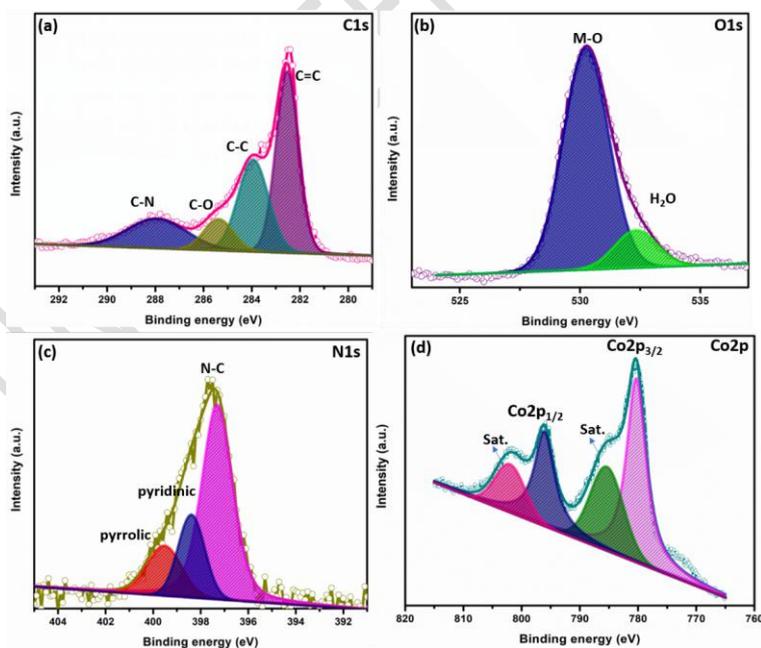


Figure 2: Scanning electron microscopy (SEM) images of (a, b) Co-MOF/CNF composite, (c) Co-MOF, and (d) CNF. (e–h) Energy-dispersive X-ray spectroscopy (EDS) elemental mapping of the Co-MOF/CNF composite, and (i) corresponding EDS spectrum

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High-resolution scanning electron microscopy (SEM) was employed to investigate the microstructure and surface characteristics of the Co-MOF/CNF composite. The SEM images of pristine Co-MOF (Figure 2a) illustrate its distinct polyhedral morphology. In contrast, Figure 2b displays the microstructure of CNF at a magnification of 1 μm , highlighting its fibrous network. The SEM analysis of the Co-MOF/CNF composite (Figures 2c and 2d) reveals a complex three-dimensional polyhedral architecture integrated with a network of CNF strands. These high-resolution images provide detailed insights into the surface morphology, showcasing the attachment of Co-MOF onto the CNF framework. The micrographs also illustrate the distribution, size, and spatial arrangement of the MOF structures, offering valuable information regarding the composite's uniformity and homogeneity.

Furthermore, energy-dispersive X-ray spectroscopy (EDS) elemental mapping was conducted to confirm the elemental composition of the Co-MOF composite. The elemental maps (Figures 2e-h) display the distribution of cobalt, carbon, oxygen, and nitrogen, represented by yellow, pink, green, and orange hues, respectively. The corresponding EDS spectrum (Figure 2i) confirms the presence of these key elements, validating the successful synthesis and integration of Co-MOF and CNF within the composite structure.



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Figure 3: Deconvoluted X-ray photoelectron spectroscopy spectra of (a) C 1s, (b) O 1s, (c) N1s, and (d) Co 2p peaks

The C1s high-resolution spectrum (Figure 3a) was deconvoluted to identify various bonding environments, including C-N interactions at 288.0 eV, C-O bonds at 285.4 eV, C-C bonds at 283.9 eV, and C=C bonds at 282.5 eV [33,34]. The O1s spectrum (Figure 3b) exhibits two prominent peaks corresponding to metal-oxygen bonds and adsorbed water molecules, respectively. In the N1s spectrum (Figure 3c), three distinct peaks are observed at binding energies of 397.3 eV, 398.4 eV, and 399.5 eV. The peaks at 398.4 eV and 399.5 eV are attributed to pyridinic and pyrrolic nitrogen species, respectively [35], while the peak at 397.2 eV corresponds to sp²-hybridized nitrogen within a benzene-like structure bonded to carbon atoms [36]. The Co 2p spectrum (Figure 3d) reveals binding energy peaks at 796.1 eV and 780.4 eV, indicative of Co²⁺ and Co³⁺ oxidation states, respectively. Additionally, satellite peaks at 785.6 eV and 802.2 eV suggest minor surface oxidation of cobalt species upon exposure to the air atmosphere [37].

Oxygen evolution reaction:

The oxygen evolution reaction (OER) electrocatalytic performance of the synthesized Co-MOF/CNF was evaluated using a conventional three-electrode setup in a 1 M KOH electrolyte. A uniform catalyst slurry was applied onto a nickel foam (Ni-foam) current collector to facilitate the measurements. Linear sweep voltammetry (LSV) was employed to assess the OER activity of the Co-MOF/CNF electrocatalyst. As illustrated in Figure 4a, the Co-MOF/CNF exhibited a significantly lower overpotential of 250 mV at a current density of 10 mA/cm² compared to Co-MOF (300 mV), CNF (340 mV), and bare Ni-foam (390 mV), demonstrating superior OER activity. The Tafel slope, which reflects the electrocatalytic kinetics during the OER process, revealed that a lower value corresponds to enhanced catalytic efficiency.

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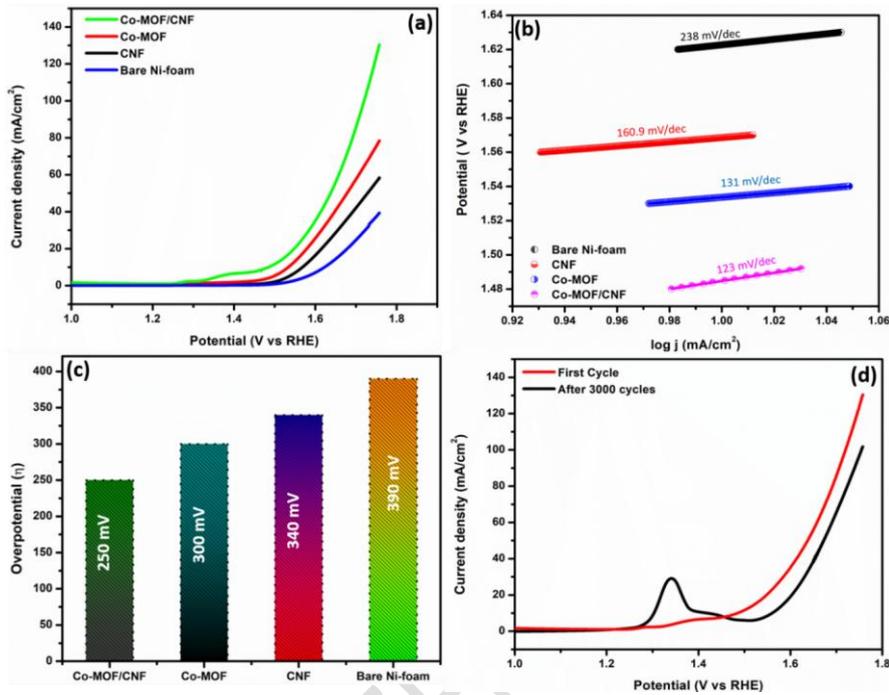


Figure 4: (a) Linear sweep voltammetry (LSV) polarization curves recorded in 1 M KOH at a scan rate of 10 mV/s for Co-MOF/CNF, Co-MOF, CNF, and bare Ni-foam electrodes; (b) corresponding Tafel plots; (c) comparison of overpotentials for the synthesized catalysts; and (d) LSV curves of Co-MOF/CNF before and after durability testing over 3000 cyclic voltammetry (CV) cycles at 50 mV/s

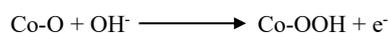
Compared to the Tafel slopes of Co-MOF (131 mV/dec), CNF (160.9 mV/dec), and bare Ni-foam (238 mV/dec) electrodes, the Co-MOF/CNF electrode exhibits a significantly lower Tafel slope, indicating superior kinetics conducive to the oxygen evolution reaction (OER), as illustrated in Figure 4b. The Co-MOF/CNF composite demonstrates both a remarkably low overpotential and a reduced Tafel slope, signifying enhanced OER efficiency. The oxygen evolution overpotentials (η) at a current density of 10 mA/cm² for Co-MOF/CNF, Co-MOF, CNF, and bare Ni-foam electrodes were determined, as presented in Figure 4c.

The enhanced electrochemical performance is attributed to the synergistic interaction between Co-MOF and CNF within the composite. The incorporation of cobalt within the imidazole framework introduces porosity, which enhances the catalyst's surface area and facilitates charge

transfer. Additionally, the nitrogen atoms embedded in the MOF framework act as secondary active sites for electrocatalysis. Due to nitrogen's higher electronegativity, it modifies the charge densities of adjacent carbon atoms, thereby lowering the adsorption free energy ($\Delta G(H^*)$) and promoting the adsorption of OER intermediates, ultimately enhancing catalytic activity [38].

Considering that OER involves a complex four-electron/four-proton transfer process, it generally requires a higher overpotential compared to the hydrogen evolution reaction (HER). Moreover, most OER catalysts derived from earth-abundant materials exhibit limited stability in acidic environments. Based on these considerations, alkaline media are deemed more favorable for achieving higher overall water-splitting efficiency compared to acidic or neutral conditions.

A proposed mechanism for the OER in an alkaline medium is outlined as follows:



Another contributing factor to the enhanced electrocatalytic performance is the nickel foam substrate. Its macroporous architecture and excellent electrical conductivity facilitate efficient charge transfer across the electrode and promote effective mass transport of electrolytes and the generated gas bubbles (H_2 and O_2) [39].

Moreover, XPS analysis confirms the presence of graphitic, pyridinic, and metal-coordinated nitrogen species within the Co-MOF/CNF catalyst, which are recognized as additional active sites for the oxygen evolution reaction (OER). As illustrated in Figure 4d, the linear sweep voltammetry (LSV) curve obtained after 3000 cycles of cyclic voltammetry at a scan rate of 50 mV/s exhibits noticeable changes compared to the initial data, likely due to the oxidation of nickel. Despite these changes, the Co-MOF/CNF catalyst demonstrates exceptional stability, with its overpotential shifting slightly from an onset potential of 1.47 V to 1.53 V (vs. RHE) after 3000 cycles.

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Hydrogen evolution reaction:

The electrocatalytic HER behaviour of Co-MOF, CNF, Co-MOF/CNF, and bare Ni-foam was investigated in a typical three-electrode setup with a scan rate of 10 mV/s in 1 M KOH solution.

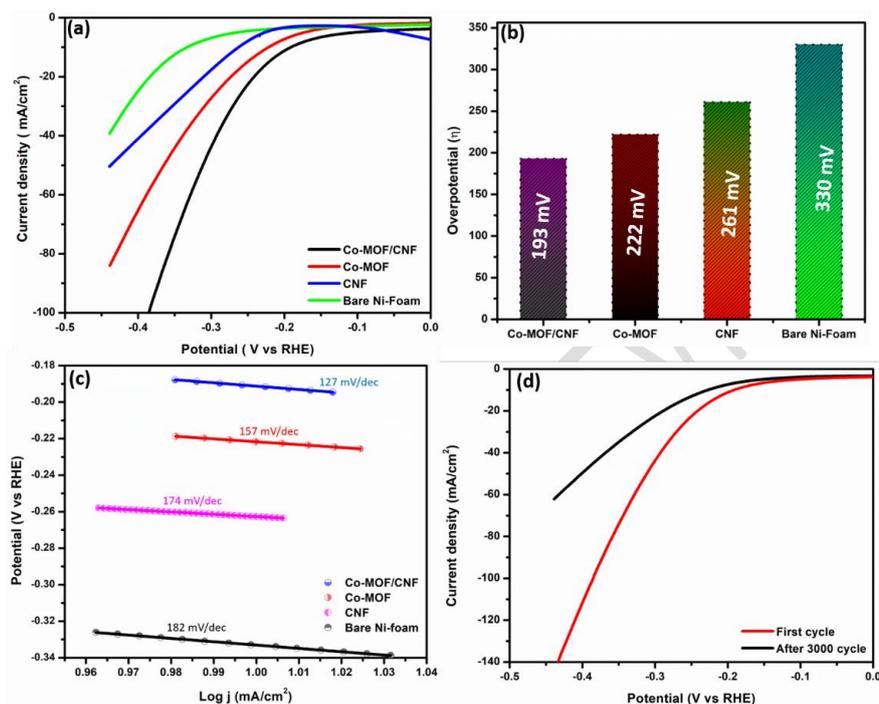


Figure 5: (a) Linear sweep voltammetry (LSV) polarization curves measured in 1 M KOH at a scan rate of 10 mV/s, (b) comparison of overpotentials among various catalysts, and (c) LSV curves obtained before and after durability testing over 3000 cyclic voltammetry (CV) cycles at 50 mV/s.

The Co-MOF/CNF composite was deposited onto nickel foam and utilized as the working electrode. As shown in Figure 5a, the LSV curves reveal that Co-MOF/CNF requires an overpotential of 193 mV to achieve a stable current density of 10 mA/cm². The overpotentials for Co-MOF, CNF, and bare Ni-foam were found to be 222, 261, and 330 mV, respectively. Figure 5b presents the computed hydrogen evolution overpotentials (η) at a current density of 10 mA/cm² for the Co-MOF/CNF, Co-MOF, CNF, and bare Ni-foam electrodes. It is believed that the synergistic interaction between nitrogen and cobalt species optimizes the electronic structure of carbon and enhances the energy of hydrogen atom adsorption on carbon, thereby

improving catalysis and electron transfer during the HER process. The corresponding Tafel plots in Figure 5c, generated by linear fitting, were used to evaluate the HER kinetics of Co-MOF, Co-MOF/CNF, CNF, and bare Ni-foam. The Tafel slope for Co-MOF/CNF is 127 mV/dec, which is smaller than the slopes for Co-MOF, CNF, and bare Ni-foam (157, 174, and 182 mV/dec, respectively). This suggests that Co-MOF/CNF facilitates HER via the Volmer-Heyrovsky combination mechanism, with desorption being the rate-determining step. In alkaline solutions, the Volmer step involves water dissociation and subsequent hydrogen reduction and adsorption ($\text{H}_2\text{O} + \text{e}^- \rightarrow \text{H}_{\text{ads}} + \text{OH}^-$), followed by either the Heyrovsky step ($\text{H}_{\text{ads}} + \text{H}_2\text{O} + \text{e}^- \rightarrow \text{H}_2 + \text{OH}^-$) or the Tafel step ($2\text{H}_{\text{ads}} \rightarrow \text{H}_2$). These results underscore the superior catalytic performance of Co-MOF/CNF, demonstrating that structural engineering can enhance both electron transfer efficiency and active site accessibility, which are critical for improving electrocatalytic activity. Stability, which refers to both material durability and sustained hydrogen evolution activity, was evaluated through prolonged CV cycling at a scan rate of 50 mV/s for 3000 cycles. The Co-MOF/CNF composite exhibited minimal deviation in the LSV curves (Figure 5d), indicating excellent stability.

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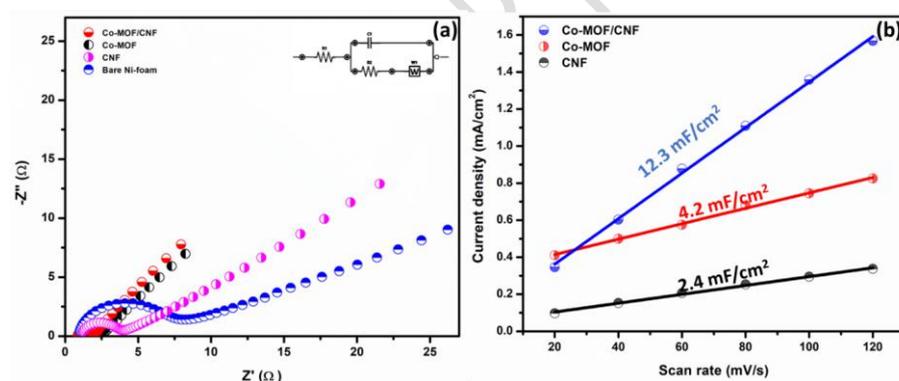


Figure 6: (a) Electrochemical impedance spectroscopy (EIS) spectra of Co-MOF/CNF, Co-MOF, CNF, and bare Ni-foam electrodes, and (b) plot of Δj (current density difference) as a function of varying scan rates.

Electrochemical impedance spectroscopy (EIS) measurements were conducted in 1 M KOH using both Co-MOF and Co-MOF/CNF electrodes within the frequency range of 100 kHz to 0.1 Hz, as depicted in Figure 6a. A lower charge transfer resistance (R_{ct}) is indicative of enhanced water-splitting performance, as it facilitates faster charge transfer and more favorable reaction kinetics [42]. The Nyquist plot reveals that Co-MOF/CNF exhibits an R_{ct} of 2 Ω ,

which is lower than that of Co-MOF (2.37 Ω), CNF (4.8 Ω), and bare Ni-foam (9.6 Ω). The smallest semicircle diameter observed for Co-MOF/CNF signifies the highest electron transfer rate among the catalysts during the OER process, directly contributing to superior catalytic activity. The incorporation of carbon nanofibers significantly improves the composite's conductivity, thereby accelerating electron migration. These findings suggest that the integration of CNF into the Co-MOF framework positively influences electron transport properties.

Additionally, the electrochemical surface area (ECSA) of the catalysts was determined through the calculation of electrochemical double-layer capacitance (Cdl) [43]. As expected, Co-MOF/CNF exhibits the highest capacitance, reflecting the presence of abundant active sites critical for enhanced catalytic performance. The Cdl values obtained from the slope of the linear plot of current density versus scan rate for Co-MOF, CNF, and Co-MOF/CNF are approximately 4.2, 2.4, and 12.3 mF/cm², respectively, as shown in Figure 6c. Despite a relatively lower ECSA, the Co-MOF/CNF composite demonstrates superior catalytic performance compared to Co-MOF, CNF, and bare Ni-foam, which can be attributed to the high intrinsic activity of the Co²⁺ active sites. The superior ECSA of the Co-MOF/CNF catalyst likely results from the uniform distribution of CNF across the hexagonal Co-MOF structure, effectively exposing more active sites for catalytic reactions.

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Overall water splitting:

The synthesized Co-MOF/CNF composite exhibits remarkable electrocatalytic performance for both the hydrogen evolution reaction (HER) and oxygen evolution reaction (OER), highlighting its potential as a bifunctional catalyst for efficient overall water splitting. To evaluate this, a two-electrode system was constructed using the Co-MOF/CNF composite (Figure 7). Notably, this alkaline electrolyzer achieved low operating voltages of 1.49 V for HER and 1.83 V for OER (Figure 7a), while delivering current densities of 10 and 50 mA/cm², respectively.

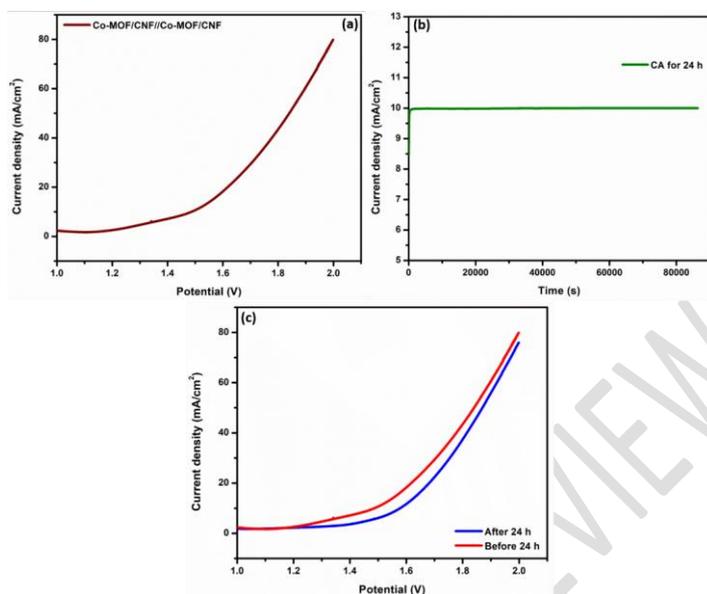


Figure 7: (a) Linear sweep voltammetry (LSV) curves obtained at a scan rate of 10 mV/s, (b) Chronoamperometry (CA) stability test performed at a current density of 10 mA/cm², and (c) LSV curves recorded after the stability test, evaluating overall water-splitting activity in an electrolysis cell with Co-MOF/CNF as both the anode and cathode in a 1 M KOH solution.

The durability of the water electrolyzer was evaluated at a current density of 10 mA/cm² (Figure 7b), with electrolysis operations sustained for over 24 hours, demonstrating the catalyst's remarkable stability for overall water splitting. Linear sweep voltammetry (LSV) results were analyzed for the composites before and after chronoamperometry, as illustrated in Figure 7c.

Based on the aforementioned evaluations and electrochemical analyses, several factors are identified as contributors to the enhanced electrocatalytic performance of the Co-MOF/CNF composite:

(i) The three-dimensional porous structure of Co-MOF offers a high density of catalytic active sites, promoting the adsorption of hydrogen- and oxygen-containing intermediates, thereby improving both hydrogen and oxygen evolution reactions.

(ii) The incorporation of CNF into the MOF alters the surface morphology of Co-MOF, which enhances the activation energy of reactants during adsorption and desorption processes on the catalyst surface, significantly boosting catalytic efficiency.

(iii) The presence of CNF within the Co-MOF catalyst matrix enhances the composite's electrical conductivity, facilitating improved energy and mass transfer at the electrode/electrolyte interface, thus accelerating the kinetics of both the hydrogen evolution reaction (HER) and oxygen evolution reaction (OER).

Conclusion:

The findings of this study emphasize the potential of three-dimensional Co-MOF/CNF microstructured assemblies as promising candidates for use in alkaline water electrolyzers. The catalyst's physical properties were thoroughly examined using appropriate analytical techniques. Additionally, electrochemical performance was assessed through various methods, including Tafel slope analysis, Electrochemical Impedance Spectroscopy (EIS), Linear Sweep Voltammetry (LSV), and Cyclic Voltammetry (CV). These analyses provided a comprehensive evaluation of the catalyst's structural and electrochemical characteristics. The results demonstrate that the Co-MOF/CNF composite exhibits superior performance in an alkaline environment, with lower overpotentials of 250 mV for the OER and 193 mV for the HER. The combination of CNF and Co-MOF offers a large active surface area, which enhances charge transfer and catalytic stability. Furthermore, the catalyst's activity was evaluated for overall water splitting, where the Co-MOF/CNF composite achieved remarkable performance, delivering 1.49 V at a current density of 10 mA/cm² in alkaline conditions. These findings highlight the critical role of synergistic material integration in achieving enhanced electrocatalytic performance and advancing water-splitting technologies. This research opens new avenues for the development of innovative catalytic materials, with significant implications for addressing global energy and environmental challenges as the search for cleaner and more efficient energy conversion solutions continues.

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